tively the safest lasers on manufacturing desktops. Since DLs are mass producible like computer chips, they will also be the cheapest lasers to buy and maintain during the next decade and will replace the current inefficient, bulky and/or environmentally unsafe lasers wherever the DL wavelengths, from 0.4 μ m to 5 μ m, are acceptable. Current high-power diodes are mostly confined to a range of around $0.8 \mu m$ and $0.98 \mu m$ for historical reasons and also because of demand from communication markets. Since these wavelengths are shorter than those of $CO₂$ (10.6 μ m) and Nd-YAG (1.06 μ m), the light energy couples more efficiently on metal work pieces. In the future, as the laser manufacturing market opens up, high-power diodes will become available over a

of them are combined to produce tens to thousands of watts coupling of such a bar of laser array to a multimode optical fiber using

LASER DESKTOP MACHINING

Harnessing a massless beam of light energy for heavy manufacturing applications like cutting, drilling, welding, soldering, surface hardening, cladding, sintering metal powder for rapid part fabrication, and so on, is almost like science fiction becoming reality to engineers (1). There is no need to replace expensive cutting tools repeatedly. The reality was demonexpensive cutting tools repeatedly. The reality was definite-
strated in the 1960s with the invention of ruby, Nd–YAG, and $CO₂$ lasers delivering pulsed high peak power. By the early 1990s, engineers realized the potential of the desktop manufacturing revolution, with the advent of compact highpower diode lasers (DL) arrays (2–17).

Of all the lasers, DLs are the most efficient ones. Some of the commercial ones now reach 40% to 50% electrical to optical efficiency. They are very compact: a 1 W laser is the size of a grain of table salt, with emission intensity about 1 $MW/cm²$ to 10 MW/cm². They operate at about 2 V, and their pulse rate and shape can be controlled to almost any desired
value from direct current (dc) to multigigahertz. DLs have not
only helped usher in the Knowledge Age through fiber-optic
communication network, but they are als of continuous-wave (CW) or pulsed power. DLs are compara- a cylindrical lens.

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Figure 2. Intensities and interaction times required for various laser material processes. (Courtesy of Lawrence Livermore National Laboratory and Fraunhofer Resource Center at Ann Arbor; see also Ref. 10.)

much wider range of wavelengths from deep blue (0.35 μ m) to near-infrared (5 μ m).

Diode lasers (Fabry–Perot type) are produced by cleaving epi-
taxially grown $p-n$ junction semiconductor materials GaAlAs
(0.8 μ m), GaAsSb (0.98 μ m), InGaAsP (1.55 μ m), and so on,
Much higher total power in the k m), GaAsSb $(0.98 \mu m)$, InGaAsP $(1.55 \mu m)$ and then electrically pumping across the $p-n$ junction orthographic mercially available with a geometry called "rack-n-stack." In onal to the microscopic waveguide cavity (5) (500 μ m to 1000 this geometry a two-dimensi μ m long) [see Fig. 1(a)]. Single-mode waveguide facets are

A monolithic array $(6-9)$ of such lasers can easily emit 1 by miniature cylindrical lenses (monolithic or discrete), fol-
W to 20 W as shown in Fig. 1(b). The light is generally inco-
herent from element to element, but collect it into a glass fiber of approximately 200 μ m core diameter (as an example), matching the width of one monolithic diode array [Fig. 1(c)]. Such a system, with 80% coupling efficiency, can easily provide an output working intensity exceeding 50 kW/cm2 .

Let us focus briefly on the intensity (or power density) requirements for various laser materials processing functions (10,11) as shown in Fig. 2. The figure does not explicitly identify processes like metal powder sintering for rapid prototyping, soldering, cellulose cutting, vapor phase deposition, and so on, that require power density in the low kilowatt domain with total power from 10 W to 100 W only. Figure 2 does show that most of the traditional laser material processing functions can (eventually) be carried out by DLs since their intrinsic emission intensity is about 107 W/cm2 albeit at low individual device power of about 1 W or less. Thus, heavy manufacturing requires collecting laser light from a large number of devices into a small spot to achieve high total **Figure 3.** High brightness diode laser spot from an incoherent bar
power at high intensity. Technically, this is called high after vertically stacking horizontal s brightness requirement; and the associated engineering diffi- and translated mirrors. (Courtesy of Opto Power Corporation.)

culty, along with the limited market acceptance, has kept the cost of the current commercial system relatively high. A diode coupled fiber bundle as shown in Fig. 1(c) is capable of providing CW power exceeding 100 W or more. Such devices are commercially available from several international suppliers. **HIGH-POWER DIODE LASERS** Figure 3 shows a different geometry to achieve intensity reaching 50 kW/cm² to 100 kW/cm². This is achieved by fold-

this geometry a two-dimensional dense stack of incoherent di- μ m long) [see Fig. 1(a)]. Single-mode waveguide facets are ode bars are stacked on very special thin cooling plates. The about 1 μ m \times 3 μ m, through which a 10° \times 60° divergent average intensity at the surfa μ m long) [see Fig. 1(a)]. Single-mode waveguide facets are ode bars are stacked on very special thin cooling plates. The about 1 μ m \times 3 μ m, through which a $10^{\circ} \times 60^{\circ}$ divergent average intensity at the

after vertically stacking horizontal source array by a pair of tilted

 $\bigwedge \bigwedge \bigwedge \bigwedge$

Low divergence beams

Embedded Bragg grating

Figure 5. Schematic diagram of a next-generation high-power DL using broad area embedded gratings. Second-order distributed Bragg reflector is part of the laser resonator and output coupler through the surface.

Figure 4. High-power "rack-n-stack" diode array. (a) A compact rack-n-stack array can provide kilowatt range power with high intensity by using microcylindrical lens array and a big spherical lens. (b) Diode laser light emitting from such a rack-n-stack structure. (c) Optical output power versus drive current for the stack. Fifty modules with 1.5 cm InGaAs bars. (Courtesy of Lawrence Livermore National Laboratory and Fraunhofer Institute.)

Figure 6. Diode laser marking by a two-dimensional array of lasers on a plastic plate.

Figure 7. Diode laser cutting of cellulose (paper) material by a scanning laser beam.

 (b)

Figure 8. Diode laser soldering of electronic components in (a) and melting of solder in (b).

exceeding 10^5 W/cm². Figure 4(a) gives the schematic diagram of such a rack system, and Fig. $4(b)$ is a photo of such a sys-

. Figure 4(a) gives the schematic diagram tion can be very wide—for example, $15 \mu m \times 200 \mu m$ instead $m \times 3 \mu m$ as in edge-emitting Fabry–Perot laser stripe. tem showing an unfocused laser beam. Figure 4(c) shows the This broader coherent source size provides a much higher optical output power curve against the dc pump current. brightness source due to lower divergence and is capable of Figure 5 shows the geometry of the next generation DL giving an intensity of 1 $MW/cm²$ or higher at a focused that emits the light vertically through the wafer surface by spot. A two-dimensional array of such lasers will eventually virtue of second-order Bragg gratings. The emission cross sec- replace most of the current lasers for laser material pro-

Figure 9. Diode laser welding, cutting, and cladding. (a) Welding of a razor blade to a steel saw blade. (b) Cutting a razor blade. (c) Cladding a valve by stellite powder. (d) Laser-assisted (heated) machining of SiN cylinder. [Courtesy of (a) Zediker of Nuvonyx, (b) SDL Inc., (c) Fraunhofer IWS, (d) Fraunhofer Institute of Production Technology.]

Figure 10. Diode laser surface hardening of a steel tape. (a) Micro-

EXAMPLES OF DIRECT DIODE MATERIAL PROCESSING Cutting Cellulose

vironmentally risky approach of using inkjet (12). A hand- Figure 7 shows an example of cutting a paper by a computer-

Figure 12. Schematic diagram of a computerized sintering system designed for producing SFF directly from metal powder by sintering thin layers of powder, sequentially one over another.

graph of pearlite phase before laser hardening. (b) Micrograph of the held, independently addressable, high-peak power diode martensite phase after hardening by diode laser. 6 shows an elementary demonstration of marking (engraving) essing. Commercialization of such lasers is expected in the plastic (Acrylonitrite butadiene styrene) with a fiber-coupled haser array. The laser wavelength was 810 nm; power density near future. was \sim 100 W/cm².

This is another growing market because leather, cotton mate- **Marking/Engraving** rial, and paper need to be cut precisely and optimally with Laser marking is an important growing market to replace en- computer intelligence for many different product markets.

Figure 11. Micrograph of the cross section of a steel plate hardened by a diode laser. (Courtesy of Lawrence Livermore National Laboratory.)

Figure 13. Two 36-layer sintered parts fabricated from 150 μ m and 44 μ m Fe powders, shown in (a) and (b), respectively, demonstrating curl-free sintering by diode lasers.

(**b**)

RAMA DELL'A

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controlled diode laser head (fiber-coupled). The material was 70 **Soldering** lb stock black paper. The laser was a CW 980 nm diode at 6 W power level with a cutting beam spot size of 0.8 mm moving at Since small electronic components and circuit-board soldering a rate of 2.5 mm/s. This technique is also useful for laminated do not require heavy melting energy, it is possible to com-

object manufacturing (LAM) where a stack of appropriately cut bine low-duty cycle, high-peak pulsed power with low dcbiased laser power to carry out fluxless soldering to achieve an environmentally cleaner manufacturing process. Figure 8 shows soldering examples; the circuit-board soldering was done with a CW 980 nm diode at 25 W (0.6 mm spot). When soldering is carried out below 1 s exposure, there is no damage in the masking layer (11). The solder globule test [Fig. 8(b)] was done with the laser at 7 W requiring less than 2 s exposure.

Welding, Cutting, Cladding, and Machining

Welding, cutting, cladding, and so on, of metal parts and tools are heavy-duty laser manufacturing jobs. Figure 9(a) shows an example of welding steel blades using a CW 40 W (at 0.8 μ m) rack-n-stack Fabry-Perot diode laser array first collimated by a cylindrical lens array, followed by a common focusing lens. Intensity at the focus exceeds 100 kW/cm2 . Figure 9(b) is a demonstration of cutting a steel blade using rack-n-stack α -DFB laser arrays (13). Figure 9(c) shows the ongoing stellite cladding process with a 1.4 kW diode laser system. The spot size was 4×2 mm, and the processing speed was 400 mm/min. Figure 9(d) is a demonstration of laser-assisted machining of a silicon nitride ceramic that is normally not machinable. The laser power was 1.2 kW, and surface roughness of $1 \mu m$ was achieved.

Surface (Transformation) Hardening

One of our earliest experiments was surface hardening of a stainless steel ribbon because $CO₂$ (10.6 μ m) laser was not economically suitable due to its high reflectance. We carried out the hardening of a 500-series stainless steel ribbon (6 mm wide, 0.1 mm thick) with a 980 nm diode laser at a power level of 15 W on a 0.8 mm spot with 3 s exposure time. Figure 10(a) shows the pearlite phase (hardness 30 on the RC scale) before heating. Figure 10(b) shows the martensite phase after heat treatment (hardness 80 on the RC scale). An example of the hardening of a steel plate (11) by a factor of four, using laser diodes, is shown in Fig. 11.

Solid Free Form from Metal Powder (Sintering)

This section establishes the advancement of diode laser technology in solid free-form fabrication (SFF) directly from metal powders by sintering with diode laser beam without any postprocessing. This is a rapidly growing field with almost a dozen different variations in approaches (1,14,15). Commercial systems using $CO₂$ and YAG lasers are already in the market. We show a series of results that demonstrate that diode lasers are better suited for the process because of the convenience of computer control, compactness, and efficient coupling of 0.8 μ m laser energy. The schematic diagram of the experimental arrangement is shown in Fig. 12. When the commercial processes are fully matured, direct diode sintering can unleash the marketing dream of supplying customized parts deliverable overnight at almost the "mass production cost."

Noncurling. Curling of metal powder sintered parts with **EXECUTE:** CO2 and YAG lasers are quite common. We found that (1) multimode fiber (600 μ m core), when it delivers a diffuse diode laser beam, easily gives curl-free layer growth when pre- (a) Part sintered by 810 nm diode shows partial melting due to higher caution is taken to make the initial layers curl-free using absorption of laser light. (b) caution is taken to make the initial layers curl-free using absorption of laser light. (b) thermalizing substrates. The results are shown in Fig. 13. bonding due to sintering only. The laser wavelength was 810 nm delivering CW 15 W laser into a 0.7 mm spot with a scan speed of 1 mm/s.

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Figure 14. Wavelength effect in sintering when same power is used.

Figure 15. Impact of particle size on diode laser sintered part density. (a) Part made of $44 \mu m$ Fe-bronze has 80% density with structure showing partial melting. (b) Part made of 150 μ m Fe-bronze has a density of 50% only.

(**b**)

(**a**)

Figure 16. Diode laser sintered part density under N₂ and Ar atmosphere. (a) Micrograph of a part with 80% density sintered in N_2 atmosphere due to iron nitride. (b) Micrograph of a part with 70% density sintered in (\mathbf{b}) Ar atmosphere.

Wavelength Effect. Absorptivity of metal powder increases **Part Density with Particle Size.** The density and the hardwith shorter laser wavelengths. This is verified by using two ness of sintered parts are of critical importance if this method different wavelengths at 810 nm and 980 nm. The two pic- is to become a commercially acceptable process for rapid protures in Fig. 14 compare the differences in sintering effects. totyping. Figure 15 shows the intuitively obvious assumption The same 44 μ m Fe powder was used in both cases under an inert Ar atmosphere using the same total power of 15 W in a were made using identical computer-controlled processes focused spot of 750 μ m at a scan speed of 1 mm/s. The test from Fe-bronze powders of 44 μ showed that, while 980 nm just sinters the powder, 810 nm sintered and also partially melted some particles due to higher absorption. is only 50% [Fig. 15(b)] and sintered only. Laser power was

that finer particles make denser and harder parts. Two parts m and 150 μ m particle sizes. The part density made of 44 μ m powder is 80% [Fig. 15(b)] with partial melting, while that made out of 150 μ m powder

als, in the sintered part is significantly better for 44 μ m powder than for 150 μ m powder.

inside the sintering chamber can be utilized to control the part density. This was verified by fabricating sintered parts under N_2 atmosphere for Fe powder in contrast to the inert **ACKNOWLEDGMENTS** Ar gas. Sintered parts grown under otherwise identical condi-

(CHEMICAL VAPOR DEPOSITION)

Laser-assisted chemical vapor deposition is another well-es- **BIBLIOGRAPHY** tablished field (16). Figure 17 shows an almost 0.5 mm tall SiC rod grown in a chamber containing tetramethylsilane Reference 1 is a lucid and comprehensive book on the subject and (TMS at 25 torr) by focusing an 8 W diode beam (at 810 nm) is quite inspiring to read. Reference 5 is into a 0.6 mm spot on an Si wafer. This particular SiC rod physics of laser diodes, but from the viewpoint of a communication took 30 min to grow. The base of the rod is about 700 μ m.

Figure 17. Micrograph of a 700 μ m tall solid rod of SiC grown from gas phase using a focused diode laser beam inside a sealed chamber 13. A. Schoenfelder et al., 2D high brightness laser diode arrays, containing tetramethylsilane. *Conf. Proc., IEEE Lasers Electroopt. Soc.,* **2**: 480, 1997.

15 CW (at 980 nm) focused to a spot size of 0.8 mm and scan Because of compactness and wavelength diversity, complex rate of 1 mm/s. Crystollographic analysis of the microstruc- three-dimensional optoelectronic and other devices can be ture showed that the miscibility of particles, of different met- fabricated on a single substrate by changing the gas composition inside a small chamber. Spatially distributed individual laser beams with different optical frequencies and beam energies can simultaneously or sequentially develop complex **Part Density with N₂ and Ar Atmosphere.** An appropriate gas multilayer optoelectronic devices comprising dielectric, metal-
side the sintering chamber can be utilized to control the lic, and semiconductor materials.

tions showed 80% density for N_2 gas, due to nitride formation,
in contrast to 70% for inert Ar gas. The micrographs are
shown in Fig. 16(a) and Fig. 16(b), respectively. Low-melting
Pb powder is another possibility of The SiC growth experiment was carried out under DARPA **SOLID FREE FORM FROM GAS PHASE**
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is quite inspiring to read. Reference 5 is a comprehensive book on the physicist. References 6–9 give the current state of technology on highpower diode lasers. Reference 17 is a recent book on design and fabrication of high-power diode lasers.

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